# **Influence of the degree of hydrolysis of**  *poly(styrene-alt-maleic* **anhydride) on miscibility with poly(vinyl acetate)**

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The influence of the hydrolysis of anhydride groups in *poly(styrene-alt-maleic* anhydride) (PSMA) on its miscibility with poly(vinyl acetate) (PVAc) is investigated. The cloudpoint curves of these blends are determined as a function of the degree of hydrolysis. The miscibility is shown to improve with increasing number of carboxylic acid groups in PSMA. Specific interaction between the acid groups of PSMA and the ester groups of PVAc, together with the so-called copolymer effect, is put forward as an explanation for this observation.

**(Keywords: miscibility; poly(styrene-a/t-maleic anhydride); poly(viayl acetate))** 

In recent years considerable attention has been paid to copolymer of styrene and maleic anhydride, and PVAc is<br>the development of new polymer blends. From a copolymer of styrene and maleic anhydride, and PVAc is the development of new polymer blends. From a investigated by measuring these cloudpoints. By varying commercial point of view this is quite understandable, the decree of budrelysis of BSMA, the number of commercial point of view this is quite understandable, the degree of hydrolysis of PSMA, the number of since it is far more easy to prepare a new blend than to corpore is vertical which enables us to investigate since it is far more easy to prepare a new blend than to carboxyl groups is varied, which enables us to investigate<br>synthesize a completely new polymer. A commercially the influence of the enecific interaction between the synthesize a completely new polymer. A commercially the influence of the specific interaction between the acid<br>attractive polymer blend is the homogeneous blend of a groups and the serboryl groups of PVAs. However, at attractive polymer blend is the homogeneous blend of groups and the carbonyl groups of PVAc. However, at poly(styrene-alt-maleic anhydride) (PSMA) and the carpating the composition of the conclumer is also poly(styrene-*alt*-maleic annydride) (PSMA) and the same time the composition of the copolymer is also poly(vinyl acetate) (PVAc), which is known to form a varied. This will have an additional effect on the poly(vinyl acetate) (PVAC), which is known to form a varied. This will have an additional effect on the non-disintegrating hydrogel, capable of absorbing water miscibility and will be discussed in terms of the recently up to 80 times its own weight<sup>1</sup>. Heterogeneous blends of developed copolymer theory<sup>3-5</sup>.<br>these polymers do not show the same stability towards The existence of specific in these polymers do not show the same stability towards The existence of specific interactions was already water.

Miscibility between PVAc and PSMA, as in all  $20\%$  w/v solutions of these polymers in butanone. At a mixtures, requires the Gibbs free energy of mixing to be mixtures, requires the Gibbs free energy of mixing to be fixed overall polymer concentration they found an inegative. Since the entropy of mixing for high-molar-<br>negative. Since the entropy of mixing for high-molar-<br>inegat negative. Since the entropy of mixing for high-molar-<br>mass components is very small, this implies that the degree of higheligie of the approximate groups of PSMA. mass components is very small, this implies that the degree of hydrolysis of the anhydride groups of PSMA.<br>enthalpy of mixing has to be negative. In general, two This was attributed to the formation of hydrogen bonds enthalpy of mixing has to be negative. In general, two<br>basically different mechanisms can lead to exothermic involving the acid groups generated during hydrolysis of mixing. First, it can result from strong specific the anhydride groups. intermolecular interactions, like hydrogen bonding, which for instance is the case between the carbonyl group of polyesters like poly( $\varepsilon$ -caprolactone) or poly( $\beta$ propiolactone) and the hydroxyl group of polyvinyl- EXPERIMENTAL phenol. As a consequence, shifts of the  $\overline{C} = \overline{O}$  and OH *Materials*<br>vibrations in the infra-red absorption spectra of these *Materials* The PSMA sample was made by copolymerization of blends are found<sup>2</sup>. Secondly, exothermic mixing in blends The PSMA sample was made by copolymerization of consisting of one or two random copolymers can be the equal amounts of styrene and maleic amiyaride in acctone<br>at reflux temperature, using benzoyl peroxide as an result of an intramolecular 'repulsion' between the at reflux temperature, using benzoyi peroxide as an initiator. The molar mass of the sample, determined with repeating units of the copolymer(s), which outranges the initiator. The molar mass of the sample, determined with<br>a Hewlett–Packard 302B vapour-pressure osmometer,

polymer blends will phase-separate at temperatures degrees of hydrolysis were prepared by refluxing 20 wt  $\frac{1}{20}$  wt  $\frac{1}{20}$ sufficiently above the so-called lower critical solution solutions in butanone with  $2 W t \frac{1}{2}$  water for different times of reaction. The determination of the degree of temperature *(LCST)*. This is in general attended by the

INTRODUCTION occurrence of a cloudpoint. In the present paper miscibility in blends of PSMA, which is an alternating miscibility and will be discussed in terms of the recently

ater.<br>Miscibility between PVAc and PSMA, as in all  $\frac{1}{20\%}$  mentioned by Greidanus *et al.*<sup>6</sup> in a preprint dealing with involving the acid groups generated during hydrolysis of

possible presence of other unfavourable interactions<sup>3-5</sup>. a Hewlett-Packard 302B vapour-pressure osmometer,<br>According to the equation of other theory miggible was  $\overline{M}_n = 8100$ . From this polymer, samples with differen According to the equation-of-state theory, miscible was  $M_n = 8100$ . From this polymer, samples with different degrees of hydrolysis were prepared by refluxing 20 wt % hydrolysis was done by titration of maleic acid units in an \* To whom correspondence should be addressed aprotic medium according to the method of Greenhow

and Jones<sup>7,8</sup> at the Plastics and Rubber Research *Infra-red spectroscopy*<br>Institute TNO in Delft, The Netherlands. The infra-red spectra

The PVAc sample was obtained from Hoechst SP3-300 spectrophotometer, using an Apple IIe (Mowolith 20). The molar mass, determined by g.p.c. microcomputer for the data handling. The samples were (Mowolith 20). The molar mass, determined by g.p.c. microcomputer for the data handling. The samples were (Waters ALC/GPC 150C), was  $\overline{M}_w = 22,000$  and  $\overline{M}_n =$  prepared by casting thin films of the blends from 9500, using tetrahydrofuran (THF) as an eluent, relative butanone solution directly on KBr discs. to polystyrene standards. The Mark-Houwink constants used for PVAc<sup>9</sup> were  $K = 1.59 \times 10^{-2}$  cm<sup>3</sup> g<sup>-1</sup> and  $a =$ 0.70, and for  $PS^{10} K = 1.14 \times 10^{-2}$  cm<sup>3</sup> g<sup>-1</sup> and  $a = 0.72$ . RESULTS AND DISCUSSION

2) at a heating rate of  $10^{\circ}$ C min<sup>-1</sup>, using an Apple IIe 2) at a neating rate of 10 C mm<sup>-</sup>, using an Apple 11e temperature reached during the first scan was microcomputer for the data handling. Polymer blends approximately  $T + 30^{\circ}$  In this way thermal bistory is were prepared by precipitation from butanone solution<br>into hexane. The onset of the  $T_g$  jump in the second scan<br>intermediate between those of the two<br>pure components is believed to be the most unambiguous into hexane. The onset of the  $T_g$  jump in the second scan pure components is believed to be the most unambiguous was taken as the glass transition temperature.

Cloudpoints were determined on thin films, which were onset of the  $T_g$  jump vs. the composition of the blends is directly cast onto glass microscope slides from butanone plotted together with the theoretical curve accord directly cast onto glass microscope slides from butanone plotted, together with the theoretical curve according to solution, using two different methods:<br>the equation of Couchman<sup>12,13</sup>

 $(1)$  By visual inspection of blends after annealing on a hot plate (Mettler FP5) at a constant temperature, which was increased 2.5 or  $5^{\circ}$ C every hour.

using a modified Cenco-TNO light scattering polymer components,  $X$  is the weight fraction of photometer, equipped with a heating block, which could component 1 and  $\Delta C_{p,i}$  gives the value of the specific heat be controlled electrically at a heating rate of jump at the  $T_g$ . The agreement with the experimental  $0.375^{\circ}$ C min<sup>-1</sup>.

stitute TNO in Delft, The Netherlands.<br>
The infra-red spectra were recorded with a Pye Unicam<br>
The PVAc sample was obtained from Hoechst SP3-300 spectrophotometer, using an Apple He prepared by casting thin films of the blends from

*Glass transitions Figure 1a* demonstrates that, for a suitable chosen molar mass of PSMA, blends of PVAc and PSMA (degree of The glass transition temperatures were measured with<br>
a Perkin–Elmer differential scanning calorimeter (DSC-<br>
the operator of the T iumn in the scand scanning as the onset of the  $T<sub>g</sub>$  jump in the second scan. The maximum approximately  $T_e + 30^{\circ}$ C. In this way thermal history is criterion of polymer miscibility and implies that these blends of PSMA and PVAc are homogeneous within the *Cloudpoint curves*<br>
Cloudpoints were determined on thin films, which were<br>
cloudpoints were determined on thin films, which were<br>
onset of the T jumn vs the composition of the blends is the equation of Couchman<sup>12,13</sup>:

$$
\ln T_{\text{gm}} = \frac{X\Delta C_{\text{p},1} \ln T_{\text{g},1} + (1 - X)\Delta C_{\text{p},2} \ln T_{\text{g},2}}{X\Delta C_{\text{p},1} + (1 - X)\Delta C_{\text{n},2}}
$$

(2) By measuring the scattered light at an angle of  $30^\circ$ , In this equation the indices 1 and 2 indicate the two



**Figure 1** (a) Thermograms for blends of PVAc and PSMA (13% hydrolysed). (b) Glass transition temperatures vs. weight fraction of PSMA ( $X_w$ ) for blends of PVAc and PSMA. The full curve was calculated using the equation of Couchman



Figure 2 (a) Thermograms for blends of PVAc and PSMA after annealing at 220°C for 45 min. (b) Glass transition temperatures vs. weight fraction of PSMA  $(X_w)$  for blends of PVAc and PSMA after annealing at 220°C for 45 min

at  $220^{\circ}$ C for 45 min results in phase separation. This can be seen in *Figure 2a,* showing the thermograms of blends of various compositions after quenching from 220°C to<br>
room temperature. In *Figure 2b* the onset  $T_g$  values of<br>
these blends are plotted. From the fact that the *T*, values room temperature. In *Figure 2b* the onset  $T_{\rm g}$  values of  $140 \vdash$ these blends are plotted. From the fact that the  $T_{g}$  values of the pure components are found, we conclude that at  $220^{\circ}$ C a separation into phases consisting almost completely of either component has taken place. Since, as will be shown, the annealing temperature is far above the *LCST*, this is completely in line with observations on various other polymer systems, such as the much studied polystyrene/poly(vinyl methyl ether) blends<sup>14</sup>. various other polymer systems, such as the much studied polystyrene/poly(vinyl methyl ether) blends<sup>14</sup>.

To study the effect of hydrolysis of PSMA on its  $\left| \frac{B}{B} \right|$ miscibility with PVAc the intensity of the scattered light as a function of temperature was observed. The PSMA samples used were all prepared by hydrolysis of the same <sup>120</sup> sample, which means that, ignoring the minor increase due to hydrolysis, the molar mass of the PSMA samples remained the same.

In *Figure 3* the cloudpoint curves of these blends are of the low mobility of the blends at temperatures just  $0.2$ <br>above  $T_{\rm x}$ , the cloudpoints were determined using the  $X_{\rm w}$ above  $T_{g}$ , the cloudpoints were determined using the 'anneal' method (see 'Experimental' section). In *Figure 4* **Figure 3** Cloudpoint curves for blends of PVAc and PSMA (X<sub>w</sub> = the cloudpoints determined for blends containing 10 wt % weight fraction of PSMA) as a function o PSMA, which is near the minima of the cloudpoint curves A, 1.8%; B, 2.1%; C, 8.2%; D, 11.4%; E, 13.0%; F, 14.2-43.0% of *Figure 3,* are plotted as a function of the degree of hydrolysis. They were measured using an extremely low approximations for the location of the thermodynamic heating rate of 0.375°C min<sup>-1</sup>. Cooling these samples phase-separation temperatures.<br>using the same low rate resulted in remixing of the blends *Figures 3* and 4 show that up to a certain point the using the same low rate resulted in remixing of the blends at approximately the same temperature, which indicates miscibility between PVAc and PSMA increases with the that the measured cloudpoints are fairly good number of acid groups in the blends. However, above



weight fraction of PSMA) as a function of the percentage of hydrolysis:



percentage of hydrolysis  $(H)$  seems at least to contain the basic ingredients  $3-5,15,16$ . In



Figure 5 Infra-red absorption spectra of a blend of PVAc with 40 wt  $\%$ Figure 5 Initia-lea absorption spectra of a biend of FVAC with 40 wt /<sub>0</sub><br>PSMA (percentage of hydrolysis = 30 %) after annealing at 110°C (A),<br>120°C (B) and 130°C (C), respectively, for 45 min<br>discussion with respect to t

15 wt% hydrolysis the cloudpoints occur at the same warranted here because of the limited amount of temperature, a somewhat unexpected result which we experimental data available. think is due to the re-formation of anhydride groups In terms of the repeating units  $\chi_{\text{eff}}$  is given by:<br>forward in hydrology DSMA at higher temperatures favoured in hydrolysed PSMA at higher temperatures. Apparently, under the given conditions this backreaction takes place at about 130°C and as a consequence the miscibility between hydrolysed PSMA and PVAc does not improve anymore.

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The number of acid and anhydride groups can be determined by studying the infra-red spectra of the blends. In *Figure 5* the infra-red spectra in the region  $2000-1500$  cm<sup> $-1$ </sup> are plotted for a blend of PVAc and 40 wt  $\%$  PSMA (percentage of hydrolysis = 30  $\%$ ) which respectively. During the annealing process the number of acid groups reduces, decreasing the absorption at 1700-  $1620 \text{ cm}^{-1}$ , while the number of anhydride groups increases, enhancing the absorptions at 1770 and ester carbonyl group did not change, providing an opportunity to use this peak for scaling of the plots. *Figure 5* clearly shows that a re-formation of anhydride groups takes place at temperatures between 110 and 130°C. From all these observations we conclude that the 110 miscibility of hydrolysed PSMA and PVAc improves as a function of the number of hydrolysed anhydride groups.

Recently, considerable progress has been made in  $\begin{array}{ccc} \hline \downarrow & \downarrow & \downarrow \\ 10 & 20 & 30 & 40 \\ \hline \end{array}$  understanding the factors governing miscibility in 0 10 20 <sup>30</sup> <sup>40</sup> polymer blends involving random or alternating  $H$ (%) copolymers. Although the theoretical description of these Figure 4 Cloudpoints for blends of PVAc with 10 wt% PSMA vs. systems is still very elementary and oversimplified, it our case hydrolysed PSMA consists of three different repeating units and PVAc has one repeating unit. Let the subscripts 1–4 denote these units:

- $1 =$ styrene
- $2 =$  maleic anhydride
- $3$  = maleic acid
- $4 =$ vinyl acetate

Let  $\phi_B$  denote the volume fraction of PVAc in the blend and  $\phi_A = 1 - \phi_B$  the volume fraction of hydrolysed PSMA.  $\begin{array}{c|c}\n\hline\n\end{array}$   $\begin{array}{c|c}\n\hline\n\end{array}$  Furthermore, let  $\phi_1$  denote the volume fraction of styrene in hydrolysed PSMA and  $h$  denote the degree of hydrolysis. In that case the free energy of mixing in the familiar Flory-Huggins formulation<sup>17,18</sup> is given by:

$$
\frac{\Delta G_{\rm m}}{RT} = \frac{\phi_{\rm A}}{r_{\rm A}} \ln \phi_{\rm A} + \frac{\phi_{\rm B}}{r_{\rm B}} \ln \phi_{\rm B} + \chi_{\rm eff} \phi_{\rm A} \phi_{\rm B} \tag{1}
$$

Here  $r_A$  and  $r_B$  are proportional to the degrees of polymerization, and  $\chi_{\text{eff}}$  depends on all components involved in a way that will be discussed later. Assuming  $\chi_{\text{eff}}$  is independent of  $\phi_A$  and  $\phi_B$ , phase separation will occur as soon as:

$$
\chi_{\rm eff} > \frac{1}{2} (1/\sqrt{r_{\rm A} + 1/\sqrt{r_{\rm B}}})^2
$$
 (2)

2000 1800 1600 1600 1600 1600 Since we will use this simple description to discuss phase Wavenumber (cm<sup>-1</sup>) separation at elevated temperatures *(LCST* behaviour), here and in the following it has to be understood that  $\chi_{\text{eff}}$  is discussion with respect to the corresponding more complicated temperature dependence of  $\chi_{\rm eff}$  in terms of the equation-of-state theory<sup>19-22</sup> could be given, but is not

$$
\chi_{\text{eff}} = \phi_1 \chi_{14} + (1 - h)(1 - \phi_1) \chi_{24} + h(1 - \phi_1) \chi_{34} \n- \phi_1 (1 - \phi_1)(1 - h) \chi_{12} - \phi_1 (1 - \phi_1) h \chi_{13} \n- (1 - \phi_1)^2 h (1 - h) \chi_{23}
$$
\n(3)

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where the various  $\chi$  parameters are again free energy rather The Netherlands, for stimulating discussions and for the than interaction parameters. Because for all blends determination of the degree of hydrolysis of the PSMA investigated the values of  $r_A$  and  $r_B$  remain fixed, it is samples. reasonable to assume that improved miscibility at higher degrees of hydrolysis corresponds to smaller, and in the light of equation (2) probably more negative, values of  $\chi_{\text{eff}}$  REFERENCES at a given temperature. Equation (3) is rather complicated and given the limited amount of experimental data in Wapenaar, K. E. D., Greidanus, P. J. and Zoetewey, C. Proc. and given the limited amount of experimental data available not very suitable for a more detailed discussion. 2 Moskala, E. J., VarneU, D. F. and Coleman, M. M. *Polymer*  However, certain conclusions can be drawn by looking at  $\frac{1}{3}$ the h dependence of  $\chi_{\text{eff}}$ :  $\chi_{\text{eff}}$ :  $\chi_{\text{1093, 16, 753}}$  Kambour, R. P., Bendler, J. T. and Bopp, R. C. *Macromolecules* 

$$
\frac{d\chi_{\text{eff}}}{dh} = (1 - \phi_1)(\chi_{34} - \chi_{24}) + \phi_1(1 - \phi_1)(\chi_{12} - \chi_{13})
$$

$$
- (1 - \phi_1)^2 (1 - 2h)\chi_{23} \tag{4}
$$

The first term is expected to be negative because  $\chi_{34} < 0$  7 Siggia, S. and Hanna, J. G. 'Quantitative Organic Analysis', owing to hydrogen bonding. Moreover  $\chi_{24}$  is probably<br>positive. Since maleic anhydride is soluble in toluene, in  $\chi_{24}$  is given box  $\chi_{24}$ . If  $\chi_{25}$  is given box  $\chi_{26}$ , New York, 1979, pp. 249-251<br>contrast t positive. Since maleic anhydride is soluble in toluene, in 8<br>contrast to succipic acid, we also expect  $\chi_{12} \leq \chi_{12}$ , in 9 contrast to succinic acid, we also expect  $\chi_{12} < \chi_{13}$ , in 9 Cane, F. and Capaciolli, T. *Eur. Polym. J.* 1978, 14, 18<br>which case the second term in equation (4) is also 10 Lyngaae-Jorgensen, J. *J. Chromatogr. Sci.* 1 which case the second term in equation (4) is also  $\frac{10}{11}$ negative. Finally it is expected that  $\chi_{23}$ , although it may Blends' (Eds D. R. Paul and S. Newman), Academic Press, New be small, is positive. Therefore all three contributions to  $\gamma_{\text{ork}}$ , 1978, Vol. 1<br> $\gamma_{\text{av}}$ /dh are probably negative. Although somewhat 12 Couchman, P. R. and Karasz, F. E. Macromolecules 1978, 11, *d* $\chi_{\text{eff}}$ /dh are probably negative. Although somewhat speculative it is not unreasonable to conclude that the  $\frac{117}{13}$  Couchman, P. R. Macromolecules 1978, 11, 1156 improved miscibility as a function of the degree of  $\frac{13}{14}$ hydrolysis  $h$  is primarily due to a combination of two effects: an increase in the number of specific interactions 15 ten Brinke, G., Rubinstein, E., Karasz, F. E. and MacKnight, W.  $(x_2)$  and an enhanced conolymer effect (mainly  $x_1, y_2$ ). A J. J. Appl. Phys. 1984, 56, 2440  $(\chi_{34})$  and an enhanced copolymer effect (mainly  $\chi_{13}$ ). A J. *J. Appl. Phys.* 1984, 56, 2440<br>more detailed interpretation is only possible once the 16 Shiomi, T., Karasz, F. E. and MacKnight, W. J. Macromolecules more detailed interpretation is only possible once the various  $\chi$ -parameters involved are determined by  $\chi$  17 Flory, P. J. 'Principles of Polymer Chemistry', Cornell<br>additional experimental studies.<br>University Press, Ithaca, NY, 1953 additional experimental studies.

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